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# Hydrous albite magmas at lower crustal pressure: new results on liquidus $H_2O$ content, solubility, and $H_2O$ activity in the system NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O-NaCl at 1.0 GPa

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Abstract The system albite-H<sub>2</sub>O serves as an important model for the generation of granitic magmas, yet relatively few experimental investigations have focused on phase relations at high pressure. This study reports new experimental results, at 1.0 GPa and 690-1050 °C, on the temperature and liquid composition at vapor-saturated melting, the H<sub>2</sub>O content of undersaturated silicate liquids in equilibrium with albite, the solubility of albite in H<sub>2</sub>O-NaCl fluids immediately below the solidus, and the activity of H<sub>2</sub>O in hydrous NaAlSi<sub>3</sub>O<sub>8</sub> liquids along the liquidus. Albite melts and dissolves congruently at all temperatures and salinities. In the NaCl-absent system, the temperature of vapor-saturated melting of low albite, confirmed by X-ray diffraction, is  $695 \pm 5$  °C and the liquid composition is  $18.14 \pm 1.35$  wt% H<sub>2</sub>O. The temperature dependence of the fluid-undersaturated liquidus curve in the system NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O varies with H<sub>2</sub>O wt% ( $w_{H_2O}$ ) according to

$$T = -2.0331 \times 10^{-3} w_{\text{H}_2\text{O}}^3 + 1.6497 w_{\text{H}_2\text{O}}^2 - 58.963 w_{\text{H}_2\text{O}} + 1235.5 \,^{\circ}\text{C}$$

indicating positive curvature in temperature-composition coordinates and a dry melting temperature of

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A. R. Makhluf amakhluf@epss.ucla.edu ~1235 °C. At 690 °C, immediately below the solidus, albite solubility decreases drastically with NaCl content of the fluid phase, from  $8.8 \pm 0.6$  wt% in the NaCl-free fluid to <sup>5</sup>2 % at NaCl concentration of only 10 mol%. Experiments determining the activity of H<sub>2</sub>O ( $a_{H_2O}$ ) in liquids at vapor-saturated melting exploited low Cl solubility in liquids and low albite solubility in the presence of H<sub>2</sub>O–NaCl fluids. The maximum Cl content of quenched glasses, only 0.95 wt%, and very low albite solubility together make possible H<sub>2</sub>O activity measurement in melts equilibrated with NaCl–H<sub>2</sub>O solutions. When combined with activity data for H<sub>2</sub>O–NaCl fluids, experimentally determined  $a_{H_2O}$  along the liquidus is described by

$$T = -469.16a_{H_2O}^{L} {}^{\frac{1}{2}} - 93.382a_{H_2O}^{L} + 1235.5 \ ^{\circ}C.$$

At 1 GPa, H<sub>2</sub>O activities in hydrous albitic melts in H<sub>2</sub>O–NaCl fluids agree with those in the presence of H<sub>2</sub>O–CO<sub>2</sub> fluids, provided appropriate fluid mixing data are used. Our results constrain an asymmetric regular solution model for the mixing of Na<sub>1/8</sub>Al<sub>1/8</sub>Si<sub>3/8</sub>O and H<sub>2</sub>O, yielding Margules parameters of  $W_{Ab} = 25.56 \pm 0.54$  and  $W_{H_2O} = 10.50 \pm 0.74$  kJ/mol. The results imply critical mixing of liquid and vapor at ~1100 °C. Using existing speciation models, our results imply that H<sub>2</sub>O dissolved in hydrous albitic liquids is almost entirely in the form of molecular H<sub>2</sub>O (OH<sup>-</sup> concentration is negligible) at temperatures near vapor-saturated melting. The results provide a more complete experimental foundation for modeling the system NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O at high pressure.

**Keywords** Activity · Albite · Liquidus · Solubility · Speciation · Brine · Solvus · Miscibility · Granite · Magma · Margules · Supercritical fluid · Critical phenomena

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#### Introduction

The fundamental importance of  $H_2O$  in the origin of granitic magma was first shown by Goranson (1938). His melting experiments on albite (NaAlSi<sub>3</sub>O<sub>8</sub>) at elevated  $H_2O$ pressures revealed a dramatic decrease of the melting temperature from above 1100 °C to 800 °C at 0.4 GPa. Subsequently, the pressure dependence of the  $H_2O$ -saturated melting point of albite was more fully characterized, and the greatest decrease in melting temperature was shown to be near 630 °C at 1.7 GPa, where the breakdown of albite to jadeite (NaAlSi<sub>2</sub>O<sub>6</sub>) plus quartz reverses the melting trend (Boettcher and Wyllie 1969).

Because alkali feldspar is the dominant mineral component of granite, the simple system NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O serves as a useful model for understanding melting in the earth's crust. Further refinements in melting experiments included KAlSi<sub>3</sub>O<sub>8</sub> and quartz under H<sub>2</sub>O-saturated conditions (Tuttle and Bowen 1958; Luth et al. 1964). Their experiments showed that if H<sub>2</sub>O is sufficiently abundant in deep-crustal processes, H<sub>2</sub>O-rich granites can form in the temperature range thought to prevail in high-grade metamorphism, leading to the concept of anatexis as "ultra-metamorphism" (Brown and Fyfe 1970).

Another illustration of the important role of  $H_2O$  in deep-crustal processes was the demonstration by Shen and Keppler (1997) of critical mixing of liquid and vapor in the system NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O. They showed, by direct observation in an externally heated diamond pressure cell, that H<sub>2</sub>O-rich albitic liquids and NaAlSi<sub>3</sub>O<sub>8</sub>-rich aqueous fluids are miscible in all proportions at pressures in the deep-crustal range and temperatures of high-grade metamorphism (1.0–1.5 GPa and 700–1000 °C). The possibility exists of supercritical or near-critical hydrous silicate magmas as important agents of deep-crustal and upper mantle metasomatism and differentiation.

A comprehensive understanding of the system NaAl-Si<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O at elevated pressures has been inhibited by lack of adequate phase composition data, principally of H<sub>2</sub>O solubility in silicate melts and the solubility of albite and silicate melts in aqueous fluids. This has compromised analysis of the H<sub>2</sub>O budget in magma genesis and prevented a phase equilibrium link between vapor-saturated melting and critical phenomena. Figure 1 illustrates outstanding problems with reference to the system NaAl-Si<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O. Most experimental determinations of melting at high pressure have focused on dry melting (Point A) or the vapor-saturated solidus (Point B). There have been numerous measurements of the H<sub>2</sub>O concentration in the melt near B (see Silver and Stolper 1985 for a summary), but they display wide experimental scatter, chiefly due to problems of H<sub>2</sub>O exsolution in quenching from high pressure (P) and temperature (T) and problems of calibration of



Fig. 1 Schematic isobaric phase relations of the system NaAl-Si<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O, projected onto the temperature-composition plane. A, dry melting point; B, H<sub>2</sub>O-saturated melting point; C, crest of the liquid + vapor solvus; D, solubility of albite in H<sub>2</sub>O at H<sub>2</sub>O-saturated melting. The section portrays moderate to high pressures, where the maximum of the liquid + vapor loop can be expected to occur at intermediate compositions

the several methods used to analyze  $H_2O$  in the glass (drying weight loss, infrared spectroscopy, mass spectroscopy). Extrapolation of the determinations to the pressure range of deep-crustal melting and metamorphism ( $0.5 \le P \le 1.5$ GPa) is accordingly uncertain.

Experimental determination of the  $H_2O$  content of fluidundersaturated albitic liquids is even more uncertain. There have previously been no reversed measurements of liquidus compositions (Line A–B, Fig. 1). The main problem is difficulty in attaining equilibrium in unfluxed (fluid-absent) systems. Such data would be of considerable importance in evaluation of the  $H_2O$  budget in "dehydration melting" (Rutter and Wyllie 1988), where the only source of  $H_2O$  for melting is that resident in hydrous minerals such as micas.

The vapor-saturation curve (Line B–C, Fig. 1) is the link between hydrous melting, silicate solubility and possible critical mixing of silicate-rich liquid and H<sub>2</sub>O-rich vapor (Point C). Most models of H<sub>2</sub>O solubility do not anticipate critical phenomena, mainly because, in order to avoid H<sub>2</sub>O exsolution in quenched glasses, experimental measurements on which they are based were limited to small H<sub>2</sub>O contents. Paillat et al. (1992) first recognized the possible existence of supercritical fluids at high pressures in the system NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O based on improved ion-probe methods of analysis of rapidly quenched charges. They found that, at pressures greater than 0.4 GPa, H<sub>2</sub>O solubility in albitic melts is prograde; that is, it increases with increasing temperature at constant pressure. This finding contrasts with all previous results and implies that liquids may acquire enough  $H_2O$  to become supercritical mixtures with NaAlSi<sub>3</sub>O<sub>8</sub>, as is the case at about 1.0 GPa in the system SiO<sub>2</sub>-H<sub>2</sub>O (Kennedy et al. 1962; Newton and Manning 2008).

A critical point (Fig. 1, Point C) was located by Shen and Keppler (1997) at 1.0 GPa near 1100 °C and ~50 wt% H<sub>2</sub>O. Their data prove very high H<sub>2</sub>O solubility in silicate liquids. At the invariant temperature of the solidus, silicate solubility in aqueous fluids should still be substantial (Point D). This is a very important consideration-despite a few determinations of albite solubility in  $H_2O$  at high P and T (e.g., Burnham in Clark 1966; Anderson and Burnham 1983), and it has been common in thermodynamic models to neglect the composition of the vapor phase in equilibrium with the melt at the melting minimum. It can be shown that, in simple regular-solution type models, the critical temperature must approach infinity as the composition of the vapor phase tends to pure  $H_2O$ . Even in cases where albite solubility is explicitly taken into account (e.g., Zeng and Nekvasil 1996), inadequate knowledge of solubility contributes to inaccurate depiction of critical phenomena in the system. Therefore, an essential ingredient to any investigation aimed at quantitative characterization of phase relations across the full binary system is the determination of solubility of albite in H<sub>2</sub>O.

A satisfactory model of hydrous melting should be consistent with measured thermodynamic data for minerals, melts and fluids, especially in attempted extrapolations beyond experimental P-T conditions. Many highquality data exist for crystalline albite and NaAlSi<sub>3</sub>O<sub>8</sub> glass, including heat capacities (Krupka et al. 1979; Stebbins et al. 1984), equation of state data for albite and glass (Lange and Carmichael, 1990; Benusa et al. 2005; Tribaudino et al. 2010) and P-V-T measurements on H<sub>2</sub>O-bearing albitic melts (Burnham and Davis 1974). Additions of accurately determined binary melt compositions coexisting with albite or aqueous fluid over ranges of temperature and pressure would be an important complement to these data.

The present study was designed to provide some of the needed phase composition data in the system NaAl-Si<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O in a pressure–temperature region representative of melt generation in the deep crust (1.0 GPa and 690–1050 °C). The new measurements include liquidus H<sub>2</sub>O contents (Line A–B, Fig. 1), liquid composition at the vapor-saturated solidus (Point B) and albite solubility in aqueous fluid at the vapor-saturated solidus (Point D). The vapor-absent liquidus compositions were determined by reversed liquid–crystal equilibration in order to eliminate the possibility of undercooling of liquid or superheating of crystals. Melt compositions at vapor saturation were investigated by textural analysis of quenched charges, by methods developed by Burnham and Jahns (1962). This

method circumvents the problem of H<sub>2</sub>O exsolution in quenched glasses that has plagued earlier investigations. The fast-quench capability of the piston-cylinder apparatus is an additional advantage-higher H<sub>2</sub>O contents of glasses can be retrieved, in many cases, without bubble formation upon rapidly quenching. The solubility of a pure natural albite was measured at temperature near the solidus, by single-crystal weight loss and by noting incipient crystal saturation in solutions with progressively greater NaAl-Si<sub>3</sub>O<sub>8</sub> content. These methods were developed to measure solubilities of many different minerals at high temperatures and pressures (Newton and Manning 2010; Manning and Aranovich 2014). In the course of the new experiments, we found that solute NaCl greatly inhibits albite solubility at high pressure and that the Cl partitioning into coexisting albitic melts is very small. These features make possible measurement of H<sub>2</sub>O activity in the melts, since H<sub>2</sub>O activity in NaCl solutions has been determined over large ranges of temperature, pressure and composition (Aranovich and Newton 1996). The measured H<sub>2</sub>O activity data are used to calibrate a thermodynamic framework that links H<sub>2</sub>Oundersaturated melts to vapor-saturated melts and supercritical fluids. This model is, in turn, useful in evaluating H<sub>2</sub>O speciation in H<sub>2</sub>O-rich feldspathic melts, using theoretical models developed by previous workers.

#### **Experimental and analytical methods**

#### **Analytical methods**

The compositions and structural state of starting materials and run products were characterized by electron microprobe (EMP) analysis and powder X-ray diffraction (XRD). Electron microprobe analysis was conducted using a JEOL 8500 instrument. Operating conditions for the analysis of albite crystals were 15 kV, 15 nA and a 10-µm beam spot size. Counting times were 20 s on the sample and 5 s on the background for all elements except the alkali metals, which were measured for 10 s on the sample and 5 s on the background. The standards used were a Tiburon albite (for Si, Al and Na), a St. Gotthard, Switzerland adularia (for K), a Great Sitkin Island, Alaska anorthite (for Ca) and a Minas Gerais, Brazil magnetite (for Fe). For analysis of hydrous glasses, the beam was defocused to 14 µm and the current lowered to 6 nA to minimize H and Na loss. The counting time was 20 s on the sample and 5 s on the background. The same standards were used, but in addition Cl was analyzed using a PETH crystal and a synthetic barium chlorapatite standard provided by the University of Michigan.

We employed powder XRD to characterize run products and determine unit cell constants of starting and run-product albite. A PANalytic powder diffractometer and Cu K $\alpha$  ( $\lambda = 1.5408$  Å) radiation were used. Scans to check for the presence or absence of albite were conducted at  $20^{\circ} < 2\theta < 60^{\circ}$  at  $\sim 2^{\circ} 2\theta \min^{-1}$ . Scans to assess albite structural state were carried out at  $28^{\circ} \le 2\theta \le 32^{\circ}$  at  $\frac{1}{2}^{\circ} 2\theta \min^{-1}$ . The (111) peak from a high purity silicon wafer at 28.440° 2 $\theta$  was used to calibrate and index the diffraction peaks. We then calculated the 2 $\theta$  difference between the 131 and 1 $\overline{31}$  peaks ( $\Delta 2\theta_{131-1\overline{3}1}$ ) to characterize the albite structural state.

## Starting materials

Starting materials were a well-characterized natural low albite from Amelia County Court House (Virginia, USA), ultrapure water (~18 M $\Omega$  cm<sup>2</sup>/cm resistivity) and reagent grade Fischer Scientific NaCl.

The largest impurities in the albite were <0.20 wt% K<sub>2</sub>O and <0.20 wt% CaO. The  $\Delta 2\theta_{131-1\overline{3}1} = 1.076^{\circ}$  in the starting albite indicates the low structural state and is consistent with results of Newton and Smith (1967). Knowledge of the structural state of the albite starting material is important because the liquidus melting relationships will change depending on structural state (Goldsmith and Jenkins 1985a, b).

Finely ground albite was used for all experiments except for single-crystal weight loss solubility experiments in H<sub>2</sub>O–NaCl solutions. The albite was ground under acetone in an agate mortar for ~30 min until complete evaporation and then dried in an oven at 600 °C for ~30 min to volatilize any organic residue.

Platinum tubing segments of  $\sim 1.25-1.75$  cm length and 2.0 or 3.5 mm diameter were used for the experiments. The tubing segments were cleaned with acetone; then one side of each of the capsules was sealed with an oxy-acetylene flame and the whole capsule annealed.

Water mole fractions ranged from 0.035 to 0.99. The dry charge was weighed into a capsule, and then the H<sub>2</sub>O was added with a microliter syringe. A mass of H<sub>2</sub>O slightly larger than the target value was added and allowed to evaporate on the balance until the desired weight was reached. The capsule was then pinched and sealed by arc welding. Weight loss on sealing was typically less than 50 µg, all of which is attributable to platinum evaporation. The amounts of total charge ranged from ~10 to 60 mg. Weighing was carried out on a Mettler-Toledo UMX2 ultra-microbalance with a stated precision of  $2 \times 10^{-4}$  mg. Multiple weighings were made after adding each component to the capsules, and weighing errors were propagated to calculate uncertainties in concentrations and activities. Capsules were weighed before and after all runs to ensure that no water was lost during the experiment. Any experiments that lost more than ~15  $\mu$ g during the course of the experiment were discarded. It is important to note that the ability to seal accurately weighed amounts of  $H_2O$  without appreciable weight loss and to demonstrate that no loss occurs during an experiment at high *P* and *T*, is essential to accurate characterization of the key phase boundaries in Fig. 1.

#### **Piston-cylinder methods**

Experiments were conducted in a piston-cylinder apparatus of either 2.54 or 1.91-cm diameter using NaCl-graphite furnace assemblies (Manning and Boettcher 1994; Newton and Manning 2000). Pressure measurements were made using Heise Co., bourdon tube gauges. Pressure uncertainties were  $\pm 0.03$  GPa. Temperature was digitally measured and controlled to  $\pm 3^{\circ}$ C using S-type thermocouples. An all-NaCl pressure medium was used at  $T \le 900$  °C (Manning and Boettcher 1994), but higher-temperature experiments required the use of small BN discs to prevent melting of the pressure medium in the vicinity of the capsule. The BN discs were 0.15 cm in length and placed immediately above and below the capsule. In addition, BN powder was packed around capsules in these high-temperature runs. The BN-NaCl pressure medium required a pressure correction of -0.06 GPa relative to the NaCl pressure medium, based on a comparison of quartz solubility in H<sub>2</sub>O at 1.0 GPa, 950 °C, using an all-NaCl assembly (Manning 1994) with the BN-NaCl assembly.

Experiments were held at 1.0 GPa and desired temperature for 3–139 h. Run duration depended on the temperature and  $H_2O$  content. At the end of an experiment, runs were rapidly quenched to below 200 °C in ~30 s from temperatures between 690 and 1050 °C.

# Interpretation of run products

Ouenched charges were retrieved from opened capsules as glassy flat wafers in cases of substantial melting, or as loose, friable aggregates in the absence of melting. Phase relations involving albite (Ab), hydrous albitic liquid (L) and  $H_2O$ -rich fluid (also referred to as the vapor phase, V) were constrained by textural analysis of run products by binocular, petrographic and scanning electron microscopy (SEM). At 1.0 GPa, the experiments bracketed the liquidus curve at temperatures between the dry melting of albite and the L + V miscibility gap at the vapor-saturated solidus. Criteria were developed to assign results to phase stability fields of Ab + V, Ab + L, L, and L + V using a binocular microscope and/or immersion oil mounts and a petrographic microscope. In the albite-H<sub>2</sub>O system, supersolidus conditions were indicated by the presence of glass in the run products. Dispersed, small, equant albite crystals, many rhomb-shaped, reflected recrystallization of starting albite. The run product seen in Fig. 2a was interpreted to indicate that run conditions corresponded to the Ab + L stability field. At a given temperature,

increasing the H<sub>2</sub>O content yielded decreasing albite abundance in quenched charges. Run products characterized by the presence of a single homogeneous glassy phase and the absence of albite crystals were assigned to the liquid field. Experiments at vapor saturation (L + V) displayed evidence of two fluid phases in run products (Fig. 2b) and were consistent with textures observed by Burnham and Jahns (1962). The presence of a free vapor phase was indicated by pockets (void spaces) that usually formed at the interface between the capsule wall and the liquid, usually near the pinched ends of the capsule. The pockets at the top of the capsule are preserved during rapid quenching. Solutes dissolved in the vapor quench to small spheroidal balls of clear glass (Fig. 2b). As noted by Burnham and Jahns (1962), these features are evidence for the presence of a separate aqueous fluid phase coexisting with a silicate liquid.

Similar textural analysis methods were used to identify the run products from the experiments involving H<sub>2</sub>O–NaCl fluids. Subsolidus charges were lightly sintered aggregates containing large (up to  $100 \,\mu$ m) subhedral to euhedral albite crystals (Fig. 2c). Two experiments of comparatively short duration contained glassy slugs crowded with euhedral albite crystals (Fig. 2a), signaling incomplete reaction (see below). Completely melted charges at T < 1000 °C contained crystal-free, oblate glass spheroids (Fig. 2d), indicating L + V coexistence. The small but nonnegligible amount of H<sub>2</sub>O dissolved into the melt could be estimated by mass balance by weighing the glassy portion of the charge for a completely melted sample and subtracting the initial mass of albite weighed into the capsule. Since some H<sub>2</sub>O exsolves from the albitic melt upon quenching, the amount of H<sub>2</sub>O measured in the quenched glass by this procedure serves as a lower bound on the actual H<sub>2</sub>O content that was in the melt. At 1000 °C and higher, some of the fully melted charges could not be removed intact, which prevented a determination of the H<sub>2</sub>O depleted from the vapor phase.



**Fig. 2** Scanning electron photomicrographs of portions of quenched capsules and charges. **a** Backscattered electron image of textures in a run interpreted to be too short to yield complete melting. The partially melted charge contains segregated melt that formed at the top of the capsule, along with coarsely recrystallized, partially dissolved albite and possible vapor spaces. The vapor space contains halite crystals that formed during quench. Run Ab247, P = 1.0 GPa, T = 950 °C, with NaCl,  $a_{H_2O} = 0.297$ . **b** Secondary electron image of a quenched charge showing evidence for equilibrium between liquid and a coexisting vapor phase at experimental conditions. Hydrous albite liquid quenches to a bubbly glass from which minor H<sub>2</sub>O has exsolved. A fluid appears in equilibrium with the liquid phase as evi-

denced by clear glass balls that formed in the quench from a former vapor space. Run Ab27, P = 1.0 GPa, T = 800 °C, with 26.13 wt % H<sub>2</sub>O, no NaCl. **c** Backscattered electron image of an unmelted charge in which albite coexisted with H<sub>2</sub>O–NaCl fluid. Quench halite was dissolved away with water, leaving behind recrystallized euhedral crystals of albite. Run Ab248, P = 1.0 GPa, T = 950 °C, with NaCl,  $a_{H_2O} = 0.259$ . **d** Backscattered electron image showing a quenched melt spheroid quenched from melt typically observed in crystal-free H<sub>2</sub>O–NaCl liquidus experiments. High intensity (*white*) specks on the glass spheroid are halite crystals. Run Ab250, P = 1.0 GPa, T = 950 °C, with NaCl,  $a_{H_2O} = 0.311$ 

#### Solubility measurements

The solubility of albite in  $H_2O \pm NaCl$  was determined in order to constrain the isobaric-invariant composition of the  $H_2O$ -rich limb of the solvus at the solidus and its variation with  $H_2O$  activity. All experiments were conducted at 1.0 GPa and 690 °C, which is 10 °C lower than the lowest temperature at which melting was observed at this pressure.

Two types of solubility experiment were carried out. Ab<sub>1</sub> experiments yield a direct solubility determination, whereas Ab<sub>2</sub> experiments bracket solubility between upper and lower bounds by establishing the presence or absence of albite in quenched charges. The Ab<sub>1</sub> experiments determined albite solubility by measuring the weight loss of single crystals in NaCl-bearing fluids. All experiments were conducted at NaCl concentrations lower than halite saturation (Aranovich and Newton 1996). Crystals were hand picked so as to fit loosely into a 1.6-mm diameter platinum inner capsule. A weighed single crystal was loaded into the inner capsule and pinched lightly in order to allow H<sub>2</sub>O fluid flow through the inner capsule. The inner capsule was burnished at both ends, punctured with four small holes of ~0.0025 cm diameter near the ends of the capsule and then weighed. The holes provided additional flow of fluid through the inner capsule. The inner capsule was loaded into the outer capsule, and NaCl was added, followed by ultrapure water. The outer capsule was then pinched and sealed by arc welding. After the experiment was run, the pH of quenched fluids was determined immediately upon puncturing of the outer capsule. The outer capsule was then opened with a razor blade and the inner capsule removed and soaked in warm water over a hot plate while stirring with a magnetic bar in order to remove quenched salt. The capsule was subsequently dried at 110 °C and weighed. This procedure was repeated up to five times until all of the halite was dissolved, as indicated by constant weight. The weight loss of the inner capsule after the experiment corresponds to the amount of albite dissolved in the solution, and thus the solubility of albite provided that dissolution is congruent. The absence of additional crystalline phases was established by optical and SEM inspection of opened inner capsules and albite surfaces.

Ab<sub>2</sub> experiments were required for runs in pure  $H_2O$  because of much higher albite solubility. This led to the formation of deep solution channels along fractures or cleavage planes of albite crystals, causing crystals to fragment into many small pieces, some of which could escape through the capsule perforations and lead to artificially high solubility. No inner capsule was used for these experiments, but instead finely ground albite was packed directly into a Pt capsule,  $H_2O$  was added, and the capsule was then pinched and sealed as done for the other experiments. Piston-cylinder methods were the same as for the

other experiments. After opening, pH was measured with Hydrion® 1.0–12.0 pH paper. The capsules were then dried and inspected to establish the presence or absence of albite. Runs were also checked for the presence of additional phases. Using masses of albite and  $H_2O$  added to these runs, complete dissolution of albite gives a minimum solubility, whereas charges in which albite did not fully dissolve provide a maximum solubility.

# Measurement of H<sub>2</sub>O activity in albite liquids

H<sub>2</sub>O activity at the liquidus was determined by bracketing the temperature of albite melting in the presence of fluids of known H<sub>2</sub>O activity. Powdered low albite was equilibrated with a H<sub>2</sub>O-NaCl solution of a given concentration. At equilibrium, the activity of H<sub>2</sub>O in albite liquid is directly returned from the bracketed melting temperature because it must equal that in the coexisting vapor phase. We used the activity formulation of Aranovich and Newton (1996) and the H<sub>2</sub>O equation of state of Holland and Powell (1991), which are applicable provided that there is negligible Cl in the melt and negligible albite solubility in the fluid (both of which were confirmed in the course of the study; see below). These experiments follow conceptually from previous work using H<sub>2</sub>O-CO<sub>2</sub> fluids (e.g., Eggler and Kadik 1979; Bohlen et al. 1982), but exploit the very different activity-composition relations in H2O-NaCl fluids (e.g., Manning and Aranovich 2014).

#### **Experimental results**

#### Equilibrium

We used several strategies to ensure and evaluate equilibrium. For the determination of the liquidus in the Ab–  $H_2O$  system, we employed long-run durations (19–93 h) to maximize the extent of reaction during runs bracketing phase boundaries. The liquidus brackets established in this manner were then checked by conducting selected reversal experiments. These were carried out in two steps. Initially, a charge with albite and  $H_2O$  was held at *T* well above the initially determined liquidus for time sufficient to completely melt the charge. The *T* was then lowered to a value at which albite crystals were expected, and held for >24 h. Charges were then quenched and analyzed using the methods as described above. In all cases, results of reversal experiments were consistent with the bracketing experiments.

Runs dealing with the change in melting temperature as a function of NaCl content were not reversed; however, generally similar run times as the pure  $H_2O$  runs and the more reactive NaCl-bearing system are assumed sufficient for equilibrium conditions. In addition, the (Al, Si) orderdisorder state of albite was determined for representative run products. Albite quenched from temperatures higher than 850 °C showed close approach to the high structural state ( $\Delta 2\theta_{131-1\bar{3}1} \ge 1.9$ ) in accordance with Goldsmith and Jenkins (1985a, b)

Solubility experiments employed run durations that were sufficient for equilibration in previous studies of the solubility of quartz in  $H_2O$  and  $H_2O$ –NaCl (e.g., Manning 1994; Newton and Manning 2000, 2008).

# Liquidus and vapor-saturated solidus in the system $NaAlSi_3O_8-H_2O$

Results of experiments constraining the liquidus in the system NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O at 1.0 GPa conducted between 700 and 1050 °C are given in Table 1 and Fig. 3. In experiments with insufficient H<sub>2</sub>O for complete melting, albite coarsened substantially to form subhedral, rhomb-shaped crystals (Fig. 2a, c). Hydrous albite liquid in all experiments quenched to glasses that appear compositionally homogenous when observed by SEM using backscattered electrons. Quenched glasses from runs with more than about 8 wt% H<sub>2</sub>O contained microscopically detectible bubbles (Fig. 2b). Bubble density and size in quenched glasses increased with bulk H<sub>2</sub>O content. The absence of additional crystalline phases in run products confirmed congruent melting, which is consistent with previous studies.

The liquidus constraints from forward experiments were checked by reversal experiments at 800–900 °C and 8.00  $\pm$  0.75 wt% H<sub>2</sub>O, and 900–1050 °C and 7.25  $\pm$  0.75 wt% H<sub>2</sub>O. Runs Ab23, Ab234 and Ab236 indicated that final conditions were subliquidus, whereas crystal-free companion runs Ab235 and Ab237 indicated final conditions above the liquidus.

The H<sub>2</sub>O-saturated solidus was constrained to lie between 690 and 700 °C. H<sub>2</sub>O-rich experiments at 690 °C showed no evidence for melting (Table 2). In contrast, quenched melt was present in all experiments at 700 °C. Our criterion for vapor saturation at immediately supersolidus conditions is the presence of a void space in the quenched capsule with clear glassy spherules precipitated from a vapor phase (Fig. 2b). The solubility of H<sub>2</sub>O at the minimum melting temperature of 695  $\pm$  5 °C is 18.14  $\pm$  1.35 wt% H<sub>2</sub>O, as constrained by runs Ab9 (16.79 wt% H<sub>2</sub>O) and Ab25 (19.48 wt% H<sub>2</sub>O), in which run products show evidence for the presence of Ab + liquid and liquid + vapor at run conditions, respectively.

The results at 700–1050 °C indicate that at 1.0 GPa, the liquidus in the system NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O displays shallow concave-up curvature (Fig. 3). The vapor-saturated and vapor-absent melting data were fitted to a third-order polynomial. We carried out three fits to the data set, in

Table 1 Experimental results in the system  $NaAlSi_3O_8\text{--}H_2O$  at 1.0 GPa

Run	T (°C)	Time (h)	H <sub>2</sub> O (wt %)	Result
Ab9	700	68	16.79	L + Ab
Ab24	700	64	18.02	L
Ab25	700	64	19.48	L + V
Ab231	700	93	22.55	L + V
Ab5	750	24	10.93	L + Ab
Ab8	750	22	11.72	L + Ab
Ab6	750	19	12.56	L
Ab2	800	50	4.06	L + Ab
*Ab23	900	22	8.01	(L)
	800	46		L + Ab
Ab1	800	50	8.79	L + Ab
Ab260	800	48	9.68	L + Ab
Ab7	800	24	10.11	L + Ab
Ab262	800	16	10.70	L + Ab
Ab4	800	24	11.40	L
Ab3	800	24	13.35	L
Ab27	800	68	26.13	L + V
Ab11	850	23	6.44	L + Ab
Ab15	850	24	7.23	L + Ab
*Ab236	900	15	7.24	(L)
	850	27		L + Ab
*Ab237	900	15	8.73	(L)
	850	27		L
*Ab234	1050	4	6.57	(L)
	900	42		L + Ab
Ab16	900	23	6.61	L + Ab
*Ab235	1050	4	7.95	(L)
	900	42		L
Ab79	950	67	4.99	L + Ab
Ab78	950	67	6.49	L
Ab258	1050	20	3.08	L + Ab
Ab259	1050	20	4.14	L

Errors in the H<sub>2</sub>O content are in the fourth decimal place

Ab albite, L liquid, V vapor

\* Reversal runs presented on two lines, the first indicating T and time where liquid can be assumed present, indicated by (L), and the second the final T, time and result

which the intercept of  $w_{H_{2O}}$  is fixed at reported dry melting temperatures of 1195 °C (Boettcher et al. 1982), 1217 °C (Birch and LeComte 1960), or 1235 °C (Boyd and England 1963). The best fit,

$$T = -2.0331 \times 10^{-3} w_{\rm H_2O}^3 + 1.6497 w_{\rm H_2O}^2 - 58.963 w_{\rm H_2O} + 1235.5 \ ^{\circ}\text{C}$$
(1)

where *T* is in °C, and  $w_{H_2O}$  is H<sub>2</sub>O concentration in wt%, was obtained using the Boyd and England (1963) melting



**Fig. 3** Experimental results in the system NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O at 1.0 GPa. *Triangles* indicate experiments constraining the liquidus curve. *Filled symbols* indicate albite +liquid; *open symbols*, liquid only. Forward and reversed experiments are indicated, respectively, with *upward* and *downward pointing triangles*. *Open circles* indicate coexisting liquid and vapor at run conditions. The liquidus curve was calculated from Eq. 1, and the liquid + vapor was calculated with derived Margules parameters (see text)

*T* and yielded the highest  $R^2$  and a standard deviation ( $\sigma$ ) of 0.85, slightly better than results based on Birch and LeComte ( $\sigma = 0.90$ ) or Boettcher et al. 1982 ( $\sigma = 1.13$ ).

#### Albite solubility in the system NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O $\pm$ NaCl

Results of experiments on the solubility of crystalline albite in  $H_2O$  and  $H_2O$  + NaCl are given in Table 2 and Fig. 4. All experiments were conducted at 690 °C and 1.0 GPa. No additional stable crystalline phases were noted in any experiments, confirming congruent dissolution of albite. Albite solubility in H<sub>2</sub>O at 690 °C was determined to be  $8.79 \pm 0.60$  wt%. Albite solubility in H<sub>2</sub>O–NaCl solutions initially decreases very strongly with addition of small amounts of NaCl (Fig. 4); at higher NaCl concentrations, the effect of progressively greater NaCl concentrations is more subdued. Extrapolation of Ab<sub>1</sub> results to zero mole percent NaCl yields agreement with Ab<sub>2</sub> experiments. The strong inhibition of albite dissolution in saline solutions is more pronounced than found in solubility studies of other minerals, including quartz at comparable temperatures (e.g., Newton and Manning 2010; Shmulovich et al. 2001). There is however a close resemblance to the near exponential drop-off in the solubility of quartz in saline solutions just below the melting temperature at higher pressures, 900-1100 °C and 1.5-2.0 GPa (Cruz and Manning 2015).

There were minute crystals (on the order of 1  $\mu$ m) of a highly birefringent quench phase in the solubility measurements in pure H<sub>2</sub>O. These were interpreted to be sodium mica or hydromica. Importantly, larger residual crystals that would signal incongruent dissolution (e.g., paragonite or corundum) were not observed in any of the experiments. Experiments with pure H<sub>2</sub>O yielded strongly alkaline quench solutions of pH  $\geq$  12, implying that the

Run Type Time (h)  $H_2O$ NaC1 NaC1 Ab in Ab out Ab solubility Quench (mg) (mol%) (mg) (wt %) pН (mg) (mg) 3.3687 12 Ab273 26 30.8334 0.5907 0.587 6.2510 8.402 (34) Ab<sub>1</sub> 81 24.9717 1.4852 1.800 2.3215 0.8013 5.434 (10) 10 Ab267 Ab<sub>1</sub> Ab266 72 25.3026 4.4040 5.092 1.6343 0.7833 2.785 (4) 9 Ab<sub>1</sub> 7 Ab268 Ab<sub>1</sub> 92 25.2444 10.7787 11.631 1.8887 1.2497 1.743 (4) 7 Ab269 25.5106 23.4898 22.109 1.8621 1.4217 0.891(1) Ab<sub>1</sub> 48 7 Ab274 Ab<sub>1</sub> 96 24.7001 33.6767 29.592 1.4835 1.1796 0.518(2) Ab270 31.1201 3.8338 <10.9682 11-12 Ab<sub>2</sub> 68 +Ab12 Ab271 32.6197 4.6485 <12.47313 Ab<sub>2</sub> 88 +AbAb275 90 19.2869 1.9980 <9.3869 12 Ab<sub>2</sub> +Ab3.0734 12 Ab276 Ab<sub>2</sub> 88 34.4563 -Ab >8.1892

Solubility reported in weight percent as  $\frac{\Delta m_{Ab}}{\Delta m_{Ab} + m_{H_2O} + m_{NaCl}} \times 100$ 

*Run types* Ab<sub>1</sub>, single-crystal weight loss experiments; Ab<sub>2</sub>, saturated/undersaturated bracketing experiments (see text). Abbreviations +Ab albite remaining after experiment, -Ab, no albite remaining after experiment. NaCl mol% is reported for the initial solution and thus does not include dissolved albite

Uncertainties: average standard error in weighings is 0.4 µg; errors propagated to Ab solubility are given in parentheses

Table 2 Results of experiments
on albite solubility in H <sub>2</sub> O–
NaCl and pure H <sub>2</sub> O solutions at
1.0 GPa, 690 °C



**Fig. 4** Albite solubility determinations in pure H<sub>2</sub>O (*squares*) and H<sub>2</sub>O–NaCl solutions (*circles*) at 1.0 GPa and 690 °C. Results show that addition of NaCl strongly inhibits the albite dissolution in NaCl-rich brines. The *curve* was hand-drawn into fit the data. The midpoint of the Ab<sub>2</sub> experiments (see text) yields albite solubility of  $8.79 \pm 0.60$  wt% in NaCl-free H<sub>2</sub>O. Compositions calculated relative to the bulk fluid composition (H<sub>2</sub>O, NaCl and dissolved solutes; Table 2). All experiments were conducted at halite-undersaturated conditions

quenched glass contains slight excesses of Al and Si, leaving the residual quench solutions NaOH-rich. Quench pH decreases markedly with increasing NaCl mole fraction, with near-neutral quench solutions produced from high-NaCl runs. The lower solubility of albite at high  $X_{\text{NaCl}}$  evidently limits the generation of alkaline quench solutions.

#### Liquidus in the system NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O-NaCl

Results of experiments determining  $H_2O$  activity at the liquidus are given in Table 3 and Fig. 5a, b. Petrographic observation of run products confirmed congruent melting in the system albite– $H_2O$ –NaCl. Selected quenched glasses from high-NaCl experiments were analyzed by EMP and found to contain 0.73–0.95 wt% Cl (Table 3), indicating low Cl solubility in the liquid. All runs contained large mass (47–81 %) of  $H_2O$ –NaCl fluid relative to albite to ensure vapor saturation (Table 3). A high fluid-rock ratio also has the advantage that the expected mass of  $H_2O$  dissolved in albitic liquids (based on the albite– $H_2O$  experiments) represents a small fraction of the  $H_2O$  added to each run. This, combined with low Cl solubility in the melt, means that the starting  $H_2O$  and NaCl masses accurately reflect the fluid composition during an experiment.

All experimental fluids had NaCl concentrations lower than that required for halite saturation at the T of interest (Aranovich and Newton 1996).

In general, run times were sufficient to yield complete melting to L + V. However, runs Ab247 and Ab252 at 950 and 1000 °C (Table 3) contained albite crystals and glass in the quenched charges. The crystal-rich portions showed vapor pockets containing halite crystals and glassy material (Fig. 2a). Because Cl solubility in the liquid is low, these are interpreted to reflect pockets of H<sub>2</sub>O–NaCl vapor present during the experiment, from which halite crystallized upon quenching. As a test of this interpretation, a longer run was conducted at 950 °C (Ab250, 76 h), in which complete melting was observed. It was therefore assumed that runs Ab247 and Ab252 reflect supraliquidus conditions.

The structural state of run-product albite was determined by XRD. Runs Ab243 (850 °C), Ab239 (900 °C) and Ab256 (1050 °C) yielded  $\Delta 2\theta_{131-1\bar{3}1}$  values of 1.909°, 1.936° and 1.952°, respectively, indicating close approach to high albite from the initially low structural state of the starting albite (Goldsmith and Jenkins 1985a, b).

Figure 5a shows the temperatures of our liquidus determinations as a function of H<sub>2</sub>O mole fraction in mixed H<sub>2</sub>O–NaCl fluids (Table 3). The data show that the temperature of vapor-saturated melting increases with decreasing H<sub>2</sub>O mole fraction in the coexisting fluid phase. In view of the slightly sigmoid dependence on H<sub>2</sub>O mole fraction, the data were fitted to a third-order polynomial with the intercept at  $X_{H_2O} = 0$  fixed at the dry melting temperature of Boyd and England (1963):

$$T = -826.52X_{\rm H_2O}^3 + 1358.0X_{\rm H_2O}^2 - 1116.7X_{\rm H_2O} + 1235.5 \ ^{\circ}{\rm C}$$
(2)

where *T* is in °C,  $X_{H_2O}$  is  $H_2O$  concentration in mole fraction, and  $R^2 = 0.997$ . Projection to dry albite melting is based on the expectation from low-pressure experiments relevant to the NaAlSi<sub>3</sub>O<sub>8</sub>–NaCl join (Kooster van Groos and Wyllie 1969), which suggest minimal melting point depression and a very albite-rich eutectic composition. This is supported by low NaCl contents of melts even at the most saline run conditions investigated yielding (<1 wt%; Table 3).

The low solubility of albite in H<sub>2</sub>O–NaCl (Fig. 4) and the low NaCl content of the hydrous albite liquids allow calculation of  $a_{H_2O}$  from the activity formulation of Aranovich and Newton (1996). Figure 5b shows the variation in the temperature of the vapor-saturated liquidus as a function of H<sub>2</sub>O activity in the liquid ( $a_{H_2O}^L$ ) determined by controlling the H<sub>2</sub>O activity in H<sub>2</sub>O–NaCl fluids ( $a_{H_2O}^V$ ) in equilibrium with hydrous silicate melt or albite. Results were extrapolated to the dry melting *T* of Boyd and England (1963) and vapor-saturated melting in the NaCl-free

Table 3Experimental constraints on liquidus relations in the system NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O-NaCl system at 1.0 GPa

Run	<i>T</i> (°C)	Time (h)	NaCl (mg)	H <sub>2</sub> O (mg)	Ab (mg)	$X_{\mathrm{H}_{2}\mathrm{O},i}$	$X_{\mathrm{H_2O},f}$	α	a <sub>H2O</sub>	Cl in glass (wt %)	Result
Ab245	800	13	19.3606	19.9008	19.7416	0.7693		1.0045	0.625		Ab + V
Ab249	800	139	10.6104	13.2025	10.8665	0.8015	0.7994	1.0045	0.665	nd	L + V
Ab246	800	50	9.5000	15.5376	10.1407	0.8414	0.8362	1.0045	0.718	nd	L + V
Ab241	850	90	20.1690	7.2503	20.5034	0.5384		0.8985	0.381		Ab + V
Ab242	850	14	20.1754	9.0180	20.5872	0.5918		0.8985	0.433		Ab + V
Ab243	850	71	20.4180	11.1416	18.9103	0.6390		0.8985	0.483		Ab + V
Ab244	850	95	21.0968	14.4060	22.4506	0.6890	0.6745	0.8985	0.522	nd	L + V
Ab239	900	68	15.4205	4.4385	21.8223	0.4829		0.7917	0.343		Ab + V
Ab240	900	69	19.8116	6.4050	20.4992	0.5119		0.7917	0.369		Ab + V
Ab238	900	16	19.9047	9.2402	20.4926	0.6010	0.5462	0.7917	0.402	0.73	L + V
Ab248	950	23	21.3362	3.8765	20.8414	0.3708		0.6851	0.259		Ab + V
Ab250	950	76	20.0090	4.9180	6.9077	0.4436	0.4316	0.6851	0.311	0.82	L + V
Ab247	950	18	18.8232	4.6488	19.8369	0.4448	0.4160	0.6851	0.297	nd	Ab + V + L
Ab253	1000	16	19.5295	2.3182	7.8092	0.2780		0.5877	0.195		Ab + V
Ab252	1000	19	19.9035	3.6788	11.6965	0.3749	_	0.5877	0.274	nd	Ab + V + L
Ab256	1050	3.3	26.6625	1.6299	11.3303	0.1655		0.4994	0.117		Ab + V
Ab297	1050	16	24.3755	1.5369	11.8917	0.1698		0.4994	0.120		Ab + V
Ab254	1050	45	25.5988	2.4034	6.5103	0.2335	_	0.4994	0.169	0.95	L + V
Ab255	1050	5	26.1134	3.1345	9.5143	0.2803	0.2647	0.4994	0.194	0.87	L + V

Ab albite, L melt, V aqueous fluid;  $X_{H_2O,i}$  initial H<sub>2</sub>O mole fraction added to capsule;  $X_{H_2O,f}$  H<sub>2</sub>O mole fraction in the vapor after correction for H<sub>2</sub>O dissolved in L (see text). with the hydrous melt ("–" where value could not be determined due to fracturing of quench glass);  $\alpha$  association parameter (Aranovich and Newton 1996);  $a_{H_2O}$ , H<sub>2</sub>O activity; *nd* not determined. Values of  $a_{H_2O}$  calculated following Aranovich and Newton (1996) using H<sub>2</sub>O density from Holland and Powell (1991) and neglecting dissolved albite in the fluid. Propagated weighing uncertainties yield an average standard error in activity of 0.007. All experiments were below halite saturation at run conditions. See text for EMP methods

system (see "Implications for albite-H2O mixing" section). Activities are on a 1-oxygen basis (i.e., a mixing unit of  $Na_{1/8}Al_{1/8}Si_{3/8}O$ ; Stolper 1982). A best fit yields

$$T = -469.16a_{\rm H_2O}^{L}{}^{\frac{1}{2}} - 93.382a_{\rm H_2O}^{L} + 1235.5\,^{\circ}{\rm C}$$
(3)

where *T* is in °C and  $R^2 = 0.996$ .

#### Discussion

# The system NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O

# $H_2O$ -saturated melting

Our results indicate that the temperature of  $H_2O$ -saturated melting is 695 ± 5 °C at 1.0 GPa. This broadly agrees with previous studies at or near this pressure, though subtle differences bear comment. Burnham and Jahns (1962), Luth et al. (1964), Boettcher and Wyllie (1969) and Bohlen et al. (1982) report wet melting temperature of <690, 692.5 ± 12.5, 685 ± 5, and 675 ± 5 °C, respectively. (The Luth et al. (1964) bracket reflects later reinterpretation by Luth (1976).) Though similar, these melting temperatures are consistently slightly lower than our value. However, all four studies employed synthetic albite crystallized from gels and relatively short-run durations. Boettcher and Wyllie (1969) found that their run-product albite was in the high structural state. The albite state of order is not reported in the other works, but use of the same starting materials and similar or shorter run durations suggests that these studies determined H2O-saturated melting of high albite. In contrast, our result reflects H2O-saturated melting of low albite, as confirmed by XRD analysis of the starting material. Goldsmith and Jenkins (1985a, b) report the melting of low albite at 700 °C and 1.1 GPa, assuming the "rare quench glass" in their run products indicates melting. They show that, at the same P, high albite + H<sub>2</sub>O yields liquid at  $685 \pm 5$  °C. Thus, the lower melting temperature of high albite likely explains the small discrepancy between our new result and earlier work. Stalder et al. (2000) reported wet melting at 1.0 GPa at about 15-20 °C higher than Goldsmith and Jenkins (1985a, b) using low albite as a starting material, but they ascribe this discrepancy to the presence of minor CO<sub>2</sub> in their charges. Notably, thermodynamic models (e.g., Silver and Stolper 1985, 1989; Zeng and Nekvasil 1996; Holland and Powell 2001) yield vaporsaturated melting temperature  $\leq 690$  °C at 1.0 GPa.



Fig. 5 Experimental results on the liquidus in the system NaAlSi $_{3}O_{8}$ -H<sub>2</sub>O-NaCl at 1.0 GPa and 695-1050 °C. a Results as a function of H<sub>2</sub>O mole fraction. Open circles indicate no melt present; filled circles, melt present (Table 3). With the exception of NaCl-free melting at H<sub>2</sub>O saturation (dotted circle), the composition of the fluid phase neglects minor dissolved albite but takes into account H2O dissolved into the melt as determined by mass balance (Table 3). The fluid composition at NaClfree vapor-saturated melting at 695 °C (*dotted circle*) is  $X_{H_2O} = 0.945$ ; dissolved albite (calculated on a 1-oxygen basis; see text) is included because of its substantially higher solubility in pure H<sub>2</sub>O (Fig. 4). The liquidus curve, derived using Eq. 2, is forced through the dry melting temperature of Boyd and England (1963), as justified by the high melting temperature at low  $X_{H_2O}$  and low solubility of Cl in the melt (see text). b Results as a function of H<sub>2</sub>O activity (Aranovich and Newton 1996), shown as *midpoints* between the most constraining experiments in a. Uncertainties in the activity determinations reflect bracket widths. As in a, fluid compositions neglect dissolved albite except for the result at H<sub>2</sub>O-saturated melting in the NaCl-free system, where  $a_{\rm H_2O} = X_{\rm H_2O} = 0.945$ . The liquidus curve was derived using Eq. 3

We find that the solubility of  $H_2O$  in  $H_2O$ -saturated albite liquid is 18.14  $\pm$  1.35 wt%  $H_2O$ . This agrees within error with previous 1.0 GPa determinations of 16.9  $\pm$  0.3 wt% (Burnham and Jahns 1962) and 18 wt% (Khitarov and Kadik 1973), which are the only previous determinations at this *P*.

Our data indicate that the solubility of albite in H<sub>2</sub>O at the vapor-saturated solidus is  $8.79 \pm 0.60$  wt% and that albite dissolves congruently at this T. The solubility is similar, but lower than the value of 10.20 wt% at 0.97 GPa and 700 °C derived by Burnham (in Clark 1966), as well as the value plotted in Fig. 1a of Khitarov and Kadik (1973). Zeng and Nekvasil (1996) provide an equation for albite solubility based on unpublished data of Paillat and Brown; however, at 690 °C and 1.0 GPa the predicted solubility of 21.15 wt% is more than a factor of two higher than our measured value, likely due to extrapolation of their model beyond conditions over which it is calibrated. Previous studies of albite solubility in H<sub>2</sub>O near 1.0 GPa and  $T \le 650$  °C demonstrated incongruent dissolution in pure H2O (Anderson and Burnham 1983; Shmulovich et al. 2001; Antignano and Manning 2008). Stalder et al. (2000) found that at 665-720 °C and 1.0 GPa, solute stoichiometry differed from that of albite, indicating incongruent dissolution. However, their fluids were interpreted to contain CO<sub>2</sub>, which is likely to modify strongly the solute stoichiometry and concentration, even at low concentration. Our finding of congruent albite dissolution about 5 °C below the solidus implies that albite stoichiometry is approached in a narrow T interval just below the solidus. This is consistent with similar results of Stalder et al. (2000) at higher P of 1.7 GPa (T = 625-690 °C) and is required by previous studies locating the vapor-saturated solidus, which point to congruent melting.

#### Liquidus and dry melting temperature

There are numerous depictions and models of the albite– $H_2O$  liquidus surface at 1.0 GPa; however,  $H_2O$ -undersaturated melting has not previously been investigated experimentally at this pressure. The experiments constrain a concave-up curvature of the liquidus (Fig. 3), which contrasts with the curvature at low pressure and with 1.0 GPa topologies shown by Khitarov and Kadik (1973) and Paillat et al. (1992) which instead show concave-down curvature. However, it is consistent with thermodynamic models (e.g., Holland and Powell 2001). Concave-up curvature is the expected topology for a system displaying closure of the L + V miscibility gap at intermediate compositions (e.g., Hack et al. 2007), as is predicted at 1.0 GPa for albite– $H_2O$  (Shen and Keppler 1997; Hayden and Manning 2011). Fig. 6 Comparison of liquidus determinations with those in previous investigations in the systems NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O-NaCl and NaAl- $Si_3O_8$ -H<sub>2</sub>O-CO<sub>2</sub> at 1.0 GPa. **a.** Results as a function H<sub>2</sub>O mole fraction. Fluid composition calculated as in Fig. 5. Open symbols, results in H2O-NaCl fluids; filled symbols, H2O-CO2 fluids. The data of Shmulovich and Graham (1996) were obtained at 0.92 GPa. The lower line is hand-drawn through the data of Bohlen et al. (1982). b Results as a function H<sub>2</sub>O activity in the liquid. All data are converted to H<sub>2</sub>O activity using the formulations of Aranovich and Newton (1996;1999). The lines represent liquidus temperature as a function of water activity in the melt from Burnham (1979), Silver and Stolper (1985), and the present study. Symbols as in a. c. Results as a function  $H_2O$  activity in the liquid, with T normalized to the melting interval used in each study. The melting interval ranges from 0 to 1 and is defined as the difference between observed T and the H<sub>2</sub>O saturated melting T, divided by the dry melting T minus the  $H_2O$  saturated melting T. Lines and symbols as in a and b

# Albite solubility and melting in the presence of mixed fluids

#### H<sub>2</sub>O-NaCl fluids

There are no previous measurements of albite solubility in H<sub>2</sub>O–NaCl fluids at 1.0 GPa. Shmulovich et al. (2001) determined albite solubility in H<sub>2</sub>O–NaCl at 0.9 GPa and 500–650 °C, but they likely encountered a field of stable paragonite at these low temperatures in pure H<sub>2</sub>O (Wohlers et al. 2011). Nevertheless, the strong decrease in albite solubility with NaCl content of the fluid at 690 °C that we observe is similar to that seen by Shmulovich et al. (2001) at 0.9 GPa and 650 °C (Fig. 6b).

Shmulovich et al. (2001) investigated the change in liquidus temperature at saturation with  $H_2O$ -NaCl fluids at 0.9 GPa and 650 °C, though over a more limited range of  $H_2O$ -rich compositions. Their results are compared with our determinations in Fig. 6a, b, where it can be seen that agreement is quite good despite the small pressure difference. In particular, our data describe a slightly sigmoidal dependence of liquidus temperature on  $H_2O$  mole fraction.

#### Comparison to $H_2O-CO_2$ fluids

Albite melting at 1.0 GPa has also been investigated in the presence of  $H_2O-CO_2$  fluid (Millhollen et al. 1971; Bohlen et al. 1982; Shmulovich and Graham 1996). Results for  $CO_2$ -bearing fluids are compared with those for NaCl-bearing fluids in Fig. 6a. The data confirm that there are large differences in the compositional dependence of the liquidus temperature when melt coexists with  $H_2O-NaCl$  versus  $H_2O-CO_2$  fluids, as noted by Shmulovich and Graham (1996).

Because NaCl and  $CO_2$  solubilities are low in hydrous albite liquid (Eggler and Kadik 1979; this study), and albite solubility is low in the two H<sub>2</sub>O bearing fluids (Shmulovich et al. 2001; this study), the difference between the liquidi in Fig. 6a results directly from the different mixing



properties of the two binary fluid systems (Manning and Aranovich 2014): Deviations from ideality are negative in the H<sub>2</sub>O–NaCl system (Aranovich and Newton 1996), but concave-up in the H<sub>2</sub>O–CO<sub>2</sub> system (Aranovich and Newton 1999). Moreover, because the melt may be considered a binary NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O mixture regardless of the coexisting mixed fluid (H<sub>2</sub>O–CO<sub>2</sub> or H<sub>2</sub>O–NaCl), the liquidus surface should be similar when portrayed as a function of H<sub>2</sub>O activity in the liquid.

Figure 6b compares liquidi in the presence of H<sub>2</sub>O–NaCl and H<sub>2</sub>O-CO<sub>2</sub> fluids. For consistency and improved accuracy, the updated activity models of Aranovich and Newton (1996, 1999) were applied to the results of this study and those of Millhollen et al. (1971), Bohlen et al. (1982), and Shmulovich and Graham (1996; data at 0.9 GPa). At lower temperatures and higher H<sub>2</sub>O concentrations, the agreement with the present data based on the activity of H<sub>2</sub>O in NaCl solutions is very good. At lower H<sub>2</sub>O concentrations and higher temperatures, the calculated liquidus H<sub>2</sub>O activity points deviate negatively from the present values. The H<sub>2</sub>O activity values shown by Silver and Stolper (1985) in their Fig. 2 are also in good agreement with the present measurements, though their activity-concentration function differs from ours. When the temperatures are scaled (Fig. 6c) according to the different dry melting temperatures of albite  $(T_w^{\circ})$  and the solidus temperatures  $(T_w)$  adopted by Millhollen et al. (1971), Bohlen et al. (1982), Shmulovich and Graham (1996) and this study, all data sets largely agree within error in the range of geologically relevant temperatures where granite genesis dominantly occurs.

Figure 6b, c shows that the liquidus H<sub>2</sub>O activity function at 1.0 GPa calculated by Burnham (1979) differs considerably from the present measurements. Using our H<sub>2</sub>O concentration in albitic liquid measurements, the activity coefficient  $\gamma_{\rm H_2O}$  ( $a_{\rm H_2O}/X_{\rm H_2O}$ ) is smaller than his estimate by a factor of nearly two at 1050 °C. In contrast, the model of Silver and Stolper (1985) agrees well with the present results.

Our results show that primary control on the albite liquidus temperature is the  $H_2O$  activity, irrespective of whether the fluid in equilibrium with the melt is composed of  $H_2O$ –  $CO_2$ , or  $H_2O$ –NaCl. This relationship was found to be true for the liquidus in the haplogranite system by Aranovich et al. (2013). This implies that our results can be used as a guide for melting in the presence of more complex fluids.

## Implications for NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O mixing

Our constraints on the compositions of  $H_2O$ -saturated albite liquid and albite-saturated vapor at the wet solidus (Fig. 1, compositions at Points B and D) allow derivation of the thermodynamic properties of liquid–vapor mixing in the albite– $H_2O$  binary. The system may be characterized using a subregular solution model, involving albite on a 1-oxygen basis (i.e., a mixing unit of  $Na_{1/8}Al_{1/8}Si_{3/8}O$ ) and  $H_2O$ . The Margules mixing parameters  $W_{Ab}$  and  $W_{H_2O}$  may be determined by setting the activities of  $H_2O$  and  $NaAl-Si_3O_8$  equal in coexisting liquid and vapor

$$a_{\rm H_2O}^L = a_{\rm H_2O}^{\rm V},$$
 (4)

$$a_{Ab}^{L} = a_{Ab}^{V},\tag{5}$$

where the activity of each component in a phase ( $\phi$ ) (i.e., the liquid or the vapor) is given by

$$\ln a_{\rm H_2O}^{\phi} = \ln X_{\rm H_2O}^{\phi} + X_{Ab}^{\phi^2} \Big[ W_{\rm H_2O} + 2X_{\rm H_2O}^{\phi} \big( W_{Ab} - W_{\rm H_2O} \big) \Big] / RT,$$
(6)

$$\ln a_{Ab}^{\phi} = \ln X_{Ab}^{\phi} + X_{H_2O}^{\phi^2} \Big[ W_{Ab} + 2X_{Ab}^{\phi} \big( W_{H_2O} - W_{Ab} \big) \Big] / RT.$$
(7)

The factor  $W_{Ab}$  in Eqs. 6 and 7 is the excess partial molar free energy change when 1/8 mol of NaAlSi<sub>3</sub>O<sub>8</sub> liquid is mixed with a large amount of H<sub>2</sub>O at fixed temperature and pressure. The two compositions input into this formulation are the albite solubility measurement in pure H<sub>2</sub>O of 8.79 ± 0.60 wt% at 690 °C and the vapor-saturated solidus composition of 18.14 ± 1.35 wt% H<sub>2</sub>O. Use of these values, in combination with Eqs. 4–7, yields Margules mixing parameters of  $W_{H_2O} = 10.50 \pm 0.74 \text{ kJ/mol}$ and  $W_{Ab} = 25.56 \pm 0.54 \text{ kJ/mol}$  and in turn a value of  $a_{H_2O}^L = a_{H_2O}^V = 0.962 \pm 0.003$  at the solidus. This compares favorably with the value of 0.945 derived from experiments by assuming that  $X_{H_2O}^V = a_{H_2O}^V$ .

iments by assuming that  $X_{H_{2O}}^V = a_{H_2O}^V$ . The value of  $a_{H_{2O}}^L = a_{H_{2O}}^V = 0.962 \pm 0.003$  at the solidus provides an anchor at NaCl-free conditions for the determination of the H<sub>2</sub>O activity along the liquidus from melting relations in H<sub>2</sub>O–NaCl fluids. In addition, accounting for the solubility of albite in H<sub>2</sub>O has important implications for the topology of the albite–H<sub>2</sub>O binary: nonzero solubility is an essential feature in a system exhibiting relatively low temperature closure of the L + V miscibility gap (Fig. 1). Although early studies recognized that the solubility of albite in H<sub>2</sub>O was finite (Burnham in Clark 1966), with the exception of Blencoe (1992) and Zeng and Nekvasil (1996), the effect has largely been ignored, presumably due to a focus primarily on phase relations in the low-H<sub>2</sub>O part of the binary.

The magnitude of the free energy change can be compared to that in other  $SiO_2$ -rich melts. Burnham and colleagues (Burnham and Davis 1974; Burnham 1975, 1979) advanced the conceptual model that melts of quartz and feldspars may be, from an energetic standpoint, considered quasi-crystalline substances. The high viscosity of the anhydrous liquids suggests that the strong Si–O bonding of the framework structure crystals is still mostly intact. Melting involves mainly loss of long-range order, which requires substantially less energy input than disruption of Si–O linkages. Thus, NaAlSi<sub>3</sub>O<sub>8</sub> liquids can, to a first approximation, be regarded as similar to a silica melt (Si<sub>4</sub>O<sub>8</sub>), on the bases of analogous crystal structures of albite and quartz, and similar responses of the liquids to dissolved H<sub>2</sub>O. Extending this analogy to dissolution behavior near the vapor-saturated solidus, if Si<sub>4</sub>O<sub>8</sub> is taken as a proxy for NaAlSi<sub>3</sub>O<sub>8</sub> and if quartz represents liquid silica, then the Gibbs free energy of the equilibrium

$$\operatorname{SiO}_{2} + 2H_{2}O = \operatorname{Si(OH)}_{4}_{\text{Solute}}$$
(8)

may be used as a comparison to  $W_{Ab}$ . Newton and Manning (2002, 2003) found that, at 1.0 GPa,

$$\Delta G_8^{\circ} = 66.84 - \frac{1.689 \times 10^4}{T} - \frac{6.520 \times 10^6}{T^2}, \qquad (9)$$

where  $\Delta G_8^\circ$  is in kJ per mole, and *T* is in Kelvin. At 900 °C,  $\Delta G_8^\circ = 47.7 \text{ kJ/mol}$  of SiO<sub>2</sub> converted to Si(OH)<sub>4</sub>, or 23.8 kJ per mole of oxygen in silica, which agrees reasonably well with 25.56 kJ for 1/8 mol of NaAlSi<sub>3</sub>O<sub>8</sub> derived from the subregular model assuming that  $W_{Ab}$  has only small variation with temperature.

Utilization of  $W_{Ab}$  and  $W_{H_2O}$  to describe the binary should account for the variation in these parameters with temperature. Although the liquidus data could be used to assess this variation, accuracy is improved using additional constraints from experimental characterization of the L + Vloop (e.g., Makhluf et al. 2013).

#### Implications for H<sub>2</sub>O speciation in albite liquids

The subregular mixing model makes no account for the speciation of  $H_2O$  in the liquid phase. Stolper (1982) showed that the primary species which must be considered are molecular  $H_2O$ and  $OH^-$ , and that their concentrations reflect the homogeneous equilibrium between three types of oxygen species in the melt

$$O^{2-} + H_2 O = 2OH^-, (10)$$

where  $O^{2-}$  represents energetically equivalent bridging oxygens. The equilibrium constant for Eq. 10 is

$$K_{10} = \frac{a_{\rm OH^-}^2}{a_{\rm H_2O} \times a_{\rm o}^{2-}}.$$
(11)

If mixing is ideal (Stolper 1982; Silver and Stolper 1985;1989),  $a_{o^{2-}}$  can be taken as  $a_{Ab}^L$ , the activity of 1/8 NaAlSi<sub>3</sub>O<sub>8</sub> (Ab) in the melt. Using equations of Stolper (1982) and Silver and Stolper (1985) along with thermodynamic data for dry albite melting from Holland and Powell (2001) and our data on liquid composition along the liquidus (Eq. 1) leads to values for  $K_{10}$  and allows calculation of the mole fractions of the three H<sub>2</sub>O species.



Fig. 7 Speciation of  $H_2O$  dissolved in hydrous albite liquids at 1 GPa, calculated using data Eq. 1 and the ideal solution model of Stolper (1982) and Silver and Stolper (1985). At high bulk  $H_2O$  concentrations and low temperatures approaching the vapor-saturated solidus, dissolved  $H_2O$  is dominated by molecular  $H_2O$  with only minor dissociation to  $OH^-$ 

Figure 7 illustrates the variation in the distribution of H<sub>2</sub>O and OH<sup>-</sup> along the liquidus, calculated assuming ideal mixing (Stolper 1982; Silver and Stolper 1985, 1989). Our new liquidus constraints imply that, with increasing bulk H<sub>2</sub>O content in the melt approaching the vaporsaturated solidus, dissolved OH<sup>-</sup> initially predominates at low  $H_2O$  concentration (high T). However, as  $H_2O$  concentration increases with declining temperature, dissolved molecular H<sub>2</sub>O becomes predominant and OH<sup>-</sup> negligible near the solidus. This is broadly consistent with results at lower P (e.g., Silver and Stolper 1985, 1989), despite complications due to the potential non-quenchability of the H<sub>2</sub>O speciation state (Nowak and Behrens 1995; Shen and Keppler 1995; Zhang 1999). Figure 7 shows that, given the modeling assumptions, as the H<sub>2</sub>O-saturated solidus is approached, the concentration of OH<sup>-</sup> tends rapidly toward zero and nearly all H<sub>2</sub>O is present in molecular form. This is an important consideration for understanding the very high dissolved bulk H<sub>2</sub>O, and the high H<sub>2</sub>O activities (near unity), which must be obtained for critical mixing of L and V phases to occur (see below).

# Implications for the NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O phase diagram at 1.0 GPa

Burnham (1979) and Bohlen et al. (1982) experimentally derived activity-concentration estimates that did not yield critical mixing of albitic liquid and  $H_2O$ . The possibility

of this important aspect of the phase relations at elevated pressures was predicted by Paillat et al. (1992) based on improved measurement of the H2O contents of glasses quenched from experiments in the binary system at pressures below 0.7 GPa. In contrast to previous workers, they found that at pressures above about 0.4 GPa, the solubility of H<sub>2</sub>O in albite melts (vapor saturation) begins to increase with increasing temperature. Paillat et al. (1992) realized that this trend would lead to critical mixing of NaAlSi<sub>3</sub>O<sub>8</sub> and H<sub>2</sub>O at high pressures; that is, that the isobaric twophase envelope between silicate melt (i.e., "liquid") and aqueous fluid (i.e., "vapor") will close to a critical point at some high temperature. They suggested a temperature of about 1250 °C at 1.0 GPa based on extrapolation of their data. The data of Shen and Keppler (1997) indicate a critical temperature near 1000 °C at 1.0 GPa, though pressure uncertainty is large. The results of Hayden and Manning (2011) and Kawamoto et al. (2012) suggested a slightly lower temperature (closer to 900 °C) of critical mixing at this pressure

The present data support critical mixing of the system NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O at 1.0 GPa. The mixing parameters  $W_{Ab} = 25.56$  kJ/mol and  $W_{H_2O} = 10.50$  kJ/mol derived from our solidus composition measurements at 690 °C and 1.0 GPa should have only small temperature dependence, since the coexisting fluids are dense and have relatively small expansivity at high pressures. Assuming that  $W_{Ab}$  and  $W_{H_2O}$  are independent of temperature in the range 700-1100 °C, the two-phase envelope is predicted to close with increasing temperature to a critical point near 1100 °C and a critical composition of 55 wt% H<sub>2</sub>O (Fig. 8a).

Figure 8b compares 1 GPa phase relations in the NaAl-Si<sub>3</sub>O<sub>8</sub>-rich part of the binary, as determined in this study and in selected previous work (Silver and Stolper 1985; Zeng and Nekvasil 1996; Holland and Powell 2001). It can be seen that the new results indicate the highest H<sub>2</sub>O solubility at the vapor-saturated solidus, consistent with direct experimental determinations (Burnham and Jahns 1962; Khitarov and Kadik 1973; this work). The decrease in liquidus T with H<sub>2</sub>O at low H<sub>2</sub>O contents is similar and would be more so if all used the same dry melting temperature of albite. Departures at higher H<sub>2</sub>O contents result from different H<sub>2</sub>O solubilities at minimum melting and different ways of treating  $K_{10}$ . The greatest difference is seen in the curve depicting the solubility of H<sub>2</sub>O at vapor saturation. Nearly vertical slopes were obtained by Silver and Stolper (1985) and Holland and Powell (2001) because they assumed that (1) H<sub>2</sub>O solubility in the liquid was independent of T and (2) H<sub>2</sub>O activity in the vapor phase at the solidus was unity. As noted above, such models cannot yield critical mixing. Zeng and Nekvasil (1996) accounted for both factors, and their model yields critical mixing. Although 1 GPa is the extreme upper pressure limit of their model, it is nevertheless apparent that predicted liquidus topology, compositions at the solidus, solvus topology and critical temperature are inaccurate. It is evident that accurate account the nonzero solubility of NaAlSi<sub>3</sub>O<sub>8</sub> in H<sub>2</sub>O at the solidus temperature of 695 °C is necessary to model phase relations in the high-H<sub>2</sub>O part of the binary.



**Fig. 8 a** Quantitative  $T - X_{H_2O}$  binary for the system NaAlSi<sub>3</sub>O<sub>8</sub>– H<sub>2</sub>O at 1.0 GPa, as constrained by results of this study. Our results predict critical mixing at a temperature of about 1100 °C and composition of ~ 55 wt % H<sub>2</sub>O. **b** Comparison of results of the present study with previous work, at relatively low bulk H<sub>2</sub>O contents. Relative to previous estimates, our new experimental data imply higher temperature of melting at H<sub>2</sub>O saturation, higher H<sub>2</sub>O solubility at vapor-saturated melting and a shallow positive slop of the albite-rich limb of the L + V solvus

#### Application to petrology

The system NaAlSi<sub>3</sub>O<sub>8</sub>–H<sub>2</sub>O is, by itself, limited in direct application to magmatic processes, though the principles it illustrates are very important to a thermodynamicsbased understanding of granites. A more general application to petrology awaits expansion of the system to include quartz and K-feldspar. A reconnaissance study of a simple ternary granite by Bureau and Keppler (1999) showed that this more complex system also displays critical mixing of melt and aqueous fluid in much the same P-T range as for albite–H<sub>2</sub>O. Further investigations involving solubility and H<sub>2</sub>O activity measurements are therefore highly desirable.

Many features of granite petrology have been linked to high mineral solubility in the vapor phase at elevated Pand T. These include such phenomena as ocellar, orbicular, graphic and rapakivi textures (Ballhaus et al. 2015). The assumption of essentially unit H<sub>2</sub>O activity at deep-crustal pressures and temperatures adopted by numerous workers should give way to considerations of high solubility of mineral components in near-critical aqueous solutions. Aranovich et al. (2013) showed that H<sub>2</sub>O activity measurements can be obtained for liquids in equilibrium with feldspars and quartz by melting experiments in the presence of (Na, K)Cl solutions. Extension of the present solubility and activity measurements to the simple granite–H<sub>2</sub>O system would seem to be feasible in developing a thermodynamic theory of granitic magmatism.

# Conclusions

- 1. Liquidus  $H_2O$  contents in the system NaAlSi<sub>3</sub>O<sub>8</sub>- $H_2O$  were determined by reversed equilibration of a pure, natural albite at 1.0 GPa and 695–1050 °C. The liquidus has concave-up curvature, with rapid increase of  $H_2O$  content near the solidus temperature of 695 ± 5 °C. The solidus  $H_2O$  content of 18.14 ± 1.35 wt% agrees well with that of Burnham and Jahns (1962) using the same method of interpretation of the run products.
- 2. Solubility measurements on albite single crystals as well as on powdered albite confirm congruent solubility and a measurement of  $8.79 \pm 0.60$  wt% albite in a pure H<sub>2</sub>O solution at 690 °C and 1.0 GPa, which is just below the melting point of albite at that pressure. This non-negligible concentration of NaAlSi<sub>3</sub>O<sub>8</sub> significantly affects the thermodynamic properties of the fluid. Critical phenomena would not be possible were it not for the appreciable solubility of albite in H<sub>2</sub>O solution. The effect of NaCl on the solubility of albite is to greatly decrease the solubility of albite in H<sub>2</sub>O rich vapor and of H<sub>2</sub>O in albite-rich silicate liquid, and in

turn, to increase the critical temperature to progressively higher temperatures not likely to be realized in typical geologic settings.

- 3. The low solubility of albite in  $H_2O$ -NaCl solutions makes possible a determination of the  $H_2O$  activity at the liquidus by equilibration of albite with concentrated NaCl solutions of known  $H_2O$  activity. The activity-concentration function along the liquidus, determined with NaCl solutions, is remarkably similar to that of Bohlen et al. (1982) when updated mixing properties of  $H_2O$ -CO<sub>2</sub> are used. This shows the power of formulating the liquidus temperature as a function of  $H_2O$  activity instead of simply  $H_2O$  concentration.
- 4. Using the  $H_2O$  speciation model of Stolper (1982) in combination with the current liquidus measurements determined here at 1.0 GPa as input show a substantial increase in molecular  $H_2O$  in the melt and a dramatic decline in  $OH^-$  near the vapor-saturated solidus. In contrast,  $OH^-$  is the dominant species at low bulk  $H_2O$ contents and high temperatures near the dry melting point of albite.
- 5. The fact that molecular H<sub>2</sub>O is the dominant species at the onset of liquid-vapor immiscibility with a value of  $a_{\rm H_2O} = 0.962 \pm 0.003$  in the melt at the vapor-saturated solidus makes possible the subregular solution model with essentially a pure end-member solution of H<sub>2</sub>O and NaAlSi<sub>3</sub>O<sub>8</sub>. As such, a Margules formulation of mixing of NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O liquid and vapor is possible with only the input of the  $T-X_{H_2O}$  values at the vaporsaturated solidus and at the invariant point on the H2Orich limb of the solvus (determined by the solubility measurement). Margules parameters of  $10.50 \pm 0.74$ and  $25.56 \pm 0.54$  kJ/mol for H<sub>2</sub>O and NaAlSi<sub>3</sub>O<sub>8</sub> (on a one-oxygen basis), respectively, were obtained at these conditions, assuming no variation of the parameters with temperature. The solvus compositions at subcritical temperatures were calculated for the NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O system with these mixing parameters. Critical phenomena are predicted at slightly higher temperatures than found by Shen and Keppler (1997; 1100 versus 1010 °C, respectively), but at the same critical composition of 55 wt% H<sub>2</sub>O at 1.0 GPa. This method may provide a viable approach to investigating the topology of the solvus of NaAlSi<sub>3</sub>O<sub>8</sub>-H<sub>2</sub>O at other pressures as well as in other mineral-H<sub>2</sub>O systems that congruently dissolve at temperatures slightly below the solidus.

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